## A Simple Approach for the Fabrication of Na<sub>1+x</sub>V<sub>3</sub>O<sub>8</sub> Nanoribbon Thin Film

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 $Na_{1+x}V_3O_8$  nanoribbon thin films have been obtained through a thermal treatment of the  $V_2O_5$  xerogel thin films prepared on sodium silicate glass substrates which also serve as a kind of sodium source at 450 °C for a certain time. The synthesized  $Na_{1+x}V_3O_8$  nanoribbon thin films were characterized by XRD, SEM, and TEM. The nanoribbons uniformly distribute on the glass substrates and have an average of several to tens of micrometers in length, hundreds of nanometers in width and tens of nanometers in thickness.

One-dimensional (1D) nanostructured materials including nanorods, nanotubes, nanowires, and nanoribbons have been considered as promising candidate materials in nanoscale electronics and photonics owing to their novel chemical and physical properties.1 Vanadium oxide bronzes have attracted many attentions because of their diverse structure and unusual electronic, photonic, and magnetic properties.<sup>2</sup> On account of their layered structures, they have been seen some potential application as cathode materials in rechargeable lithium batteries.<sup>3</sup> Most recently, a few 1D nanostructured vanadium oxide bronzes including  $Na_2V_6O_{16} \cdot 3H_2O$  nanobelts<sup>4a</sup> and nanowires,<sup>4b</sup>  $CaV_6O_{16} \cdot 3H_2O$  nanoribbons,<sup>4c</sup>  $NaV_6O_{15}$  nanowires<sup>4b</sup> and  $\beta$ -AgVO<sub>3</sub> nanowires<sup>4d</sup> have been synthesized through hydrothermal reactions. Here, we report a novel approach to synthesize  $Na_{1+x}V_3O_8$  and form nanoribbon structure by an analogous solid-state interface chemical reaction between vanadium oxide and sodium silicate glass. Along with the synthesis, these 1D ribbons were fabricated into thin films. This route is rapid, low-cost and large-scale growth of nanoribbons. The structure of  $Na_{1+x}V_3O_8$ consists of V<sub>3</sub>O<sub>8</sub> layers which are built up with VO<sub>6</sub> octahedral and V<sub>2</sub>O<sub>8</sub> units of edge-sharing VO<sub>5</sub>-VO<sub>5</sub> trigonal bipyramids and the layers are strongly held together by Na<sup>+</sup> ions residing in the octahedral sites.<sup>5</sup>

First, vanadium oxide gel was synthesized following a quenching-like route.<sup>6</sup> In a typical route, 3 g of divanadium pentaoxide powder (99%, Zhonglian Chem&Ind. Reagent Factory, Beijing) was put into a ceramic crucible and maintained at  $\approx 1000$  °C for 15 minutes, and then the molten liquid was quickly poured into 240 mL of deionized water while being stirred. The concentration of the sol solution is 0.05 M, determined through thermogravimetric analysis. If the concentration is higher than 0.1 M, gelation occurs,<sup>7</sup> and this is not benificial to synthesize nanoribbon structure. Second, a sodium silicate glass substrate has been precleaned and was dipped into the V<sub>2</sub>O<sub>5</sub> sol several times to form a xerogel film. The thickness of the film was controlled around  $1-2\,\mu m$ . Finally, the film was heated to 450 °C at a rate of 3-4 °C/min in a furnace and maintained at 450 °C for 30 min or longer. After cooling down to the room temperature slowly, a translucent brown thin film was obtained.

X-ray powder diffraction (XRD) patterns were recorded on a

Philips X' pert diffractometer with Cu K $\alpha$  radiation ( $\lambda = 1.5406$  Å). X-ray photoelectron spectroscopy (XPS) measurements were conducted on an AXIS ULTRA system (Shimadzu-Kratos Co.) using a monochromatized Al K $\alpha$  radiation as excitation source ( $h\nu = 1486.6$  eV). Scanning electron microscopy (SEM) image was taken on a JSM-5600LV scanning electron microscope. The transmission electron microscopy (TEM) images and selected area electron diffraction (SAED) pattern were obtained on a JEOL JEM-2010 transmission electron microscope at 200 kV.



**Figure 1.** (a) XRD patterns of the  $Na_{1+x}V_3O_8$  nanoribbon thin film. (b) XPS spectra of the  $Na_{1+x}V_3O_8$  nanoribbon thin film. The inset shows the  $V2p_{3/2}$  XPS core level spectra of the film.

Figure 1a shows the XRD pattern of the film obtained by thermal treated at 450 °C for 2h. All the peaks in Figure 1a can be readily indexed to a pure monoclinic crystalline phase [space group: P2<sub>1</sub>/m(11)] of Na<sub>5</sub>V<sub>12</sub>O<sub>32</sub> (JCPDS No. 24-1156, which can be considered as Na<sub>1,25</sub>V<sub>3</sub>O<sub>8</sub> compared with the cell parameters and the space hkl of  $Na_{1+x}V_3O_8$  refined by Kawakita et al.<sup>5a</sup>) with calculated lattice constants a =12.14(4) Å, b = 3.66(1) Å, c = 7.31(6) Å,  $\beta = 107.48(9)^{\circ}$ . The XRD result shows that the nanoribbons are  $Na_{1+x}V_3O_8$  with a high degree of crystallinity and the film is highly oriented for there are only some special peaks existing. The XPS spectra (Figure 1b) show that the nanoribbons consist of Na, V (in a ratio of ca. 1:2.3) and O. Furthermore, a V2p<sub>3/2</sub> binding energy of 517.2 eV is characteristic of  $V^{5+}$  and a relevant amount (ca. 10%) of  $V^{4+}$  at a characteristic  $V2p_{3/2}$  binding energy of 515.95 eV can also be seen.<sup>8</sup> Calculated from the ratio of Na, V, and the content of  $V^{4+}$ , *x*, is determined about 0.3.

Figure 2 illustrates that ribbon-like materials, with several to tens of micrometers in length, around hundreds of nanometers in width and tens of nanometers in thickness, are uniformly distributed, and the ribbons are isolated from each other with amorphous tropism. The rectangle-like cross section of the materials is clearly visible in the SEM image.

These nanoribbons can be peeled off from the glass substrate by ultrasonic and dispersed into some organic solvent. Figure 3a shows a TEM image of a nanoribbon's cleavage end. The SAED pattern (inset in Figure 3a) is well consistent with monoclinic



**Figure 2.** Typical SEM images of the  $Na_{1+x}V_3O_8$  nanoribbon thin film. (a) Low magnification; (b) High magnification.



**Figure 3.** (a) A TEM image of a nanoribbon's cleavage end. The inset shows the corresponding SAED pattern. (b) An HRTEM image from the boxed region in (a) indicates the space of 0.53(5) nm between the (101) crystal planes with the growth direction  $\langle 101 \rangle$  shown by the arrow.

phase Na<sub>5</sub>V<sub>12</sub>O<sub>32</sub> (JCPDS No. 24-1156, another chemical formula: Na<sub>1.25</sub>V<sub>3</sub>O<sub>8</sub>) and indicates that the nanoribbons grow along the [101] direction. Figure 3b is a representative enlarged HRTEM image of the boxed region in Figure 3a, which shows the clearly resolved interplanar distances  $d_{101} = 0.53(5)$  nm and  $d_{\overline{2}11} = 0.29(8)$  nm.

Further experiments have been carried out for understanding the growth mechanism. When a very thin xerogel film (dipped once) was used and heated at 450 °C, only a thin a axis-oriented NaVO<sub>3</sub> film could be obtained (not shown in the paper). This implies that the films must be thick enough to obtain the nanoribbons and one layer of NaVO3 may form at the interface of vanadium oxide film and the sodium silicate glass substrate in the case of thick films. The formation of NaVO<sub>3</sub> may be similar to Mekki's reports<sup>9</sup> that Na<sup>+</sup> could be removed from the silicate network through the formation of metavanadate units of NaVO<sub>3</sub> when adding  $V_2O_5$  into the sodium silicate glasses. NaVO<sub>3</sub> can further react with vanadium oxide to form NaV<sub>3</sub>O<sub>8</sub>. And Na<sup>+</sup> ions can diffuse across the interfacial layer of NaVO3 and the upper vanadium oxide layer. On the other hand, studying the films heated at different temperature shows that the film already turns into Na<sub>1+x</sub>V<sub>3</sub>O<sub>8</sub> film treated at 350 °C for about 2 hours but no ribbon-like structure exists. Heated at 400 °C for 2 hours the film contains a few ribbon-like materials. Heated at 450 °C for 30 min large amounts of ribbon-like materials appear on the film. It may be through a vapor-solid (VS) growth mechanism that Na<sub>1+x</sub>V<sub>3</sub>O<sub>8</sub> could grow into a ribbon-like structure with high length-to-width ratio. However, heated at 450 °C for more than 24 hours only a very thin *a* axis-oriented NaVO<sub>3</sub> film is obtained. This indicates that the Na<sub>1+x</sub>V<sub>3</sub>O<sub>8</sub> nanoribbons are somewhat unstable and subliming at 450 °C. So a competitive process of subliming and depositing of Na<sub>1+x</sub>V<sub>3</sub>O<sub>8</sub> may occur on the surface of the film at high enough temperature and the nanoribbons begin to form just below at 450 °C. Further studies will be summarized in our next paper.

In summary, Na<sub>1+x</sub>V<sub>3</sub>O<sub>8</sub> nanoribbon thin films have been obtained through a thermal treatment of the vanadium oxide films prepared on sodium silicate glass substrates at 450 °C. So-dium silicate glass serves as a kind of sodium source. Through XRD, XPS, and TEM characterization the compound was identified as Na<sub>1+x</sub>V<sub>3</sub>O<sub>8</sub> of high crytallinity. The growth direction of the nanoribbon was determined and the growth mechanism was discussed.

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## References

- 1 Y. Xia, P. Yang, Y. Sun, Y. Wu, B. Mayers, B. Gates, Y. Yin, F. Kim, and H. Yan, *Adv. Mater.*, **15**, 353 (2003).
- V. Bondarenka, Z. Martunas, S. Kaciulis, and L. Pandolfi, J. Electron Spectrosc. Relat. Phenom., 131–132, 99 (2003); G. Liu and J. E. Greedan, J. Solid State Chem., 114, 499 (1995); H. Eisaki, T. Ido, K. Magoshi, M. Mochizuki, H. Yamatsu, T. Ito, and S. Uchida, Physica C, 185–189, 1295 (1991).
- 3 M. S. Whittingham, Chem. Rev., 104, 4271 (2004).
- 4 a) J. Yu, J. C. Yu, W. Ho, L. Wu, and X. Wang, J. Am. Chem. Soc., 126, 3422 (2004) b) G.-T. Zhou, X. Wang, and J. C. Yu, Cryst. Growth Des., ASAP Web Release Date: 15-Dec-2004, DOI: 10.1021/cg0496686. c) L. Kong, M. Shao, Q. Xie, J. Liu, and Y. Qian, J. Cryst. Growth, 260, 435 (2004). d) Y. Liu, Y. Zhang, Y. Hu, and Y. Qian, Chem. Lett., 34, 146 (2005).
- 5 a) J. Kawakita, K. Makino, Y. Katayama, T. Miura, and T. Kishi, *Solid State Ionics*, **99**, 165 (1997). b) K. West, B. Zachau-christiansen, and T. Jacobsen, *Solid State Ionics*, **28–30**, 1128 (1988).
- 6 E. Z. Müller, Z. Chem. Ind. Kolloide, 8, 302 (1911); D. Yin, N. Xu, J. Zhang, and X. Zheng, Mater. Res. Bull., 31, 335 (1996).
- 7 O. Pelletier, P. Davidson, C. Bourgaux, C. Coulon, S. Regnault, and J. Livage, *Langmuir*, 16, 5295 (2000).
- 8 E. Z. Kurmaev, V. M. Cherkashenko, Y. M. Yarmoshenko, S. Bartkowski, A. V. Postnikov, M. Neumann, L.-C. Duda, J. H. Guo, J. Nordgren, V. A. Perelyaev, and W. Reichelt, *J. Phys.: Condens. Matter*, **10**, 4081 (1998); M. Demeter, M. Neumann, and W. Reichelt, *Surf. Sci.*, **41**, 454 (2000).
- 9 A. Mekki, G. D. Khattak, D. Holland, M. Chinkhota, and L. E. Wenger, J. Non-Cryst. Solids, 318, 193 (2003).